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# Effect of rare-earth doping in $R\text{CrSb}_3$ ( $R = \text{La, Pr, Sm, and Gd}$ )

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## Abstract

We report on the electrical resistivity and magnetic susceptibility of La or Gd doped  $R\text{CrSb}_3$  ( $R=\text{La, Pr, Sm, and Gd}$ ). Single crystals were grown by increasing the nominal dopant by 25%. In general, two magnetic ordering transitions are found,  $T_{C1}$  is attributed to ferromagnetic ordering of the itinerant Cr sub-lattice, and, at lower temperatures,  $T_{C2}$  is attributed to ordering of the localized rare-earth sub-lattice. Alloying on the rare-earth site varies the de Gennes factor,  $DG = (g-1)^2 J(J+1)$ , and  $dT_{C1}/d(DG) = -2K$ , while  $dT_{C2}/d(DG) = 5K$ . These ordering temperatures are found to converge at  $\text{GdCrSb}_3$ , where a single ferrimagnetic transition is found at  $T_{C2}=86$  K due to an anti-alignment of the itinerant Cr moments and the localized rare-earth moments. Initially, for  $DG < 3.5$ , the rare-earth moments are found to align ferromagnetically, and the paramagnetic Weiss temperature decreases at the same rate as  $T_{C1}$ . But for  $DG > 4.5$ , the rare-earth sub-lattice anti-aligns with respect to the Cr sub-lattice, and the Weiss temperature decreases five times as fast. In the region between ( $3.5 < DG < 4.5$ ), a first order phase transition is found at  $T_{C2}$ .

## I. INTRODUCTION

The ternary compounds  $R\text{CrSb}_3$  ( $R = \text{La-Nd, Sm, Gd-Dy, Yb}$ ) have been studied extensively due to a number of their interesting physical properties. Among these are their quasi-planar crystal structure with in the  $b$ - $c$  plane as seen in Fig. 1,[1] the anomalous magnetism found in  $\text{LaCrSb}_3$ ,[2–6] and the complex interaction between the itinerant  $3d$ -electrons from the Cr ions and the localized  $4f$ -moments from the rare-earth ions.[7–13] The parent member in the series,  $\text{LaCrSb}_3$ , has been studied in depth [2–4] because it has only one magnetic ion, Cr, yet it has a complex magnetic phase diagram including a ferromagnetic (FM) transition at  $T_{C1}=132$  K, followed by a spin reorientation phase for  $T < 98$  K and  $H < 250$  G. The members  $R=\text{Ce-Nd}$ , and Sm show a second magnetic ordering at lower temperatures ( $T_{C2}=10\text{-}30$  K) due to the alignment of the rare-earth ions.[7–12] The rare-earth ordering is ferromagnetic for  $R=\text{Ce, Pr, and Nd}$ , but antiferromagnetic (AFM) and first order for  $R=\text{Sm}$ . Polycrystalline samples of  $R=\text{Gd, Tb, and Dy}$  show only one magnetic transition.[8, 11] Single crystals of  $\text{GdCrSb}_3$  indicate this is a ferrimagnetic transition at  $T_{C2}=86$  K due to anti-alignment of the Cr sub-lattice and the  $\text{Gd}^{3+}$  sub-lattice.[13] Recently, the series has been extended out to  $\text{YbCrSb}_3$ , where a single FM transition was found at 280 K, the largest magnetic ordering temperature in this family of compounds.[14] This large value was believed to be due in part to the divalent  $\text{Yb}^{2+}$  ion.

In this study, we alloy with La or Gd because they can be thought of as being on opposite ends of the spectrum and act as a convenient way to investigate the interactions between the localized rare-earth  $4f$ -moments and the itinerant Cr  $3d$ -electrons. The unit cell volume is largest for  $\text{LaCrSb}_3$  and decreases due to the lanthanide contraction going to  $\text{GdCrSb}_3$ . While they both have only one magnetic ordering temperature, the type of ordering is different for the two, and all the compounds in between have two distinct magnetically ordered phases. By doping on the rare-earth site, we can vary the number of  $4f$ -electrons which provides a means to investigate the behavior of the second magnetic ordering temperature.

We investigate  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  as a way to observe the transition from one end of the series to another, and track the evolution of the second ordering temperature from  $T_{C2}=0$  K for  $\text{LaCrSb}_3$  to  $T_{C2}=86$  K for  $\text{GdCrSb}_3$ . The first order AFM transition found in  $\text{SmCrSb}_3$  at  $T_{C2}=30$  K has been investigated in detail elsewhere (see ref. [12]) and is unique to this family of compounds, which makes it a great candidate to study.  $\text{PrCrSb}_3$  was chosen

because it has very similar properties with  $\text{NdCrSb}_3$ , [7–9] which has been investigated in detail by neutron diffraction measurements [10] showing that the Cr sub-lattice orders along the  $b$  axis at  $T_{C1}=108$  K, and the second transition at  $T_{C2}=13$  K is due to FM alignment of both the Cr and the  $\text{Nd}^{3+}$  lattices along the  $a$  axis, the stacking direction. By doping these compounds with  $\text{La}^{3+}$  (removing  $4f$  electrons), one can track the disappearance of the rare-earth ordering. On the other hand, the development of the ferrimagnetic phase can be followed by doping with  $\text{Gd}^{3+}$  (increasing  $4f$  electrons).

## II. EXPERIMENTAL TECHNIQUE

The procedure for growing single crystals of  $R\text{CrSb}_3$  has been previously described [3]. These samples crystallize in an orthorhombic structure (space group  $Pbcm$ ), [1] and consist of two distinct planar layers of Cr and Sb perpendicular to the  $a$  axis, which are separated by the rare-earth ions (Fig. 1). The first of these layers consist of chains of Cr atoms extending along the  $c$  direction, with Sb atoms forming face sharing (edge sharing) octahedra along the  $c$  axis ( $b$  axis). The rare-earth ions lie in a checkerboard-type pattern which alternates above and below the Sb plane. Due to this quasi-2D crystal structure, anisotropy plays a crucial role in understanding its properties. The lattice parameters were determined using a commercial Scintag x-ray diffractometer using a Si standard, and a least squares fit to a minimum of 20 peaks. Each alloy was grown by increasing the nominal dopant concentration by 25%, and Vegard’s Law was then used to determine the actual dopant concentration from the unit cell volume. The resulting crystals were rectangular planes with a thin face parallel to the  $a$  axis, and the longest side parallel to the  $c$  axis. The lattice constants for the materials investigated here are shown in the inset to Fig. 1.

Magnetization and magnetic susceptibility measurements were taken with a commercial superconducting quantum interference device (SQUID) magnetometer (Quantum Design MPMSR2) in the temperature range 2-350 K and the magnetic field aligned along each of the three principle axes. All samples were field cooled. Electrical resistivity measurements were performed using a standard four probe technique in the temperature range 5-295 K. All measurements were collected on as grown samples. The electrical resistivity of  $\text{SmCrSb}_3$  was investigated under pressure up to 8 kbar using a non-magnetic Be-Cu pressure clamp and flourinert as a pressure medium. The pressure was determined by the load applied to

the pressure clamp at room temperature.

### III. RESULTS

We will begin with a discussion of  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$ , because it goes directly between the endpoints in the series. Fig. 2 shows the normalized electrical resistivity as a function of temperature along the  $a$  axis (out-of-plane). For  $\text{LaCrSb}_3$ , the slope changes sign at  $T_{C1}=132$  K.[3] With the addition of  $\text{Gd}^{3+}$ , a striking upturn is observed at the rare-earth ordering temperature. This sudden increase in the electrical resistivity increases in temperature and merges with the higher temperature FM ordered phase at  $\text{GdCrSb}_3$ .

Figure 3 shows the magnetization vs temperature along the  $c$  axis (in-plane) of  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  with an applied field of 1 kG. Despite the fact that  $\text{Gd}^{3+}$  has a half-full  $4f$ -electron shell ( $\mu_{eff} = 7\mu_B$ ), and  $\text{La}^{3+}$  has no  $4f$ -electrons, we find that, due to the disappearance of the FM ground state, increasing the concentration of  $\text{Gd}^{3+}$  actually reduces the magnetization. Furthermore, the increase in magnetization at  $T_{C1}$  shifts to lower temperatures with increasing  $\text{Gd}^{3+}$ .  $\text{LaCrSb}_3$  has a shift in the magnetic easy axis from an alignment in the direction of the magnetic field within the plane immediately below  $T_{C1}$ , to alignment along the  $b$  axis as the temperature is reduced. A result of this is the drop in the magnetization below 75 K. The drop at this point seen in Fig. 3 is therefore not due to a second magnetic ordering transition, but just realignment of the easy axis. On the other hand, the peak found for  $x = 0.34, 0.74$ , and 1.0 does pinpoint the location of  $T_{C2}$ . No such peak is seen for  $x = 0.25$ , which may be due to the dominating presence of the Cr ferromagnetism at this low concentration of  $\text{Gd}^{3+}$ .

The magnetization along the  $c$  axis at 5 K for  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  is shown in Fig. 4. First, as expected for a FM ground state, the magnetization for  $\text{LaCrSb}_3$  very quickly saturates. For 25% Gd, the magnetization initially jumps up slightly, then remains constant for nearly 20 kG, followed by a linear increase with higher fields. As the  $\text{Gd}^{3+}$  concentration increases further, a met-magnetic transition develops and for  $\text{GdCrSb}_3$ , a sharp spin-flop transition is found at 25 kG. Recall that the magnitude of  $M(T)$  steadily decreases with increasing  $\text{Gd}^{3+}$ . This surprising observation can be reconciled with the  $M(H)$  results because none of the alloys with  $\text{Gd}^{3+}$  are ever found to saturate up to 55 kG, which is due to the ground state no longer being FM.

The series of alloys  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  show that the only one to have an itinerant FM ground state is  $\text{LaCrSb}_3$ . By adding  $\text{Gd}^{3+}$ , a second magnetic ordering temperature is present,  $T_{C2}$ , which is located by a kink in the electrical resistivity and a peak in the magnetization. Finally, the magnetization at a constant field dramatically decreases due to the non-FM ground state with the addition of  $\text{Gd}^{3+}$ .

$\text{SmCrSb}_3$  is unique in the family of compounds because it is found to have a strong first order phase transition at  $T_{C2}=30$  K (Fig. 5, and see ref. [12] for a detailed review). By increasing the  $\text{Gd}^{3+}$  concentration,  $T_{C2}$  increases to higher temperatures (Fig. 5(b)). While the transition is very sharp ( $\sim 4$  K wide) for  $\text{Sm}_{0.77}\text{Gd}_{0.23}\text{CrSb}_3$ , no hysteresis was found, and the transition does not appear to be first order. In addition, as was found for  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$ , the magnitude of the magnetization decreases with increasing  $\text{Gd}^{3+}$  concentration.

On the other hand, doping with  $\text{La}^{3+}$  quickly suppresses  $T_{C2}$ . With an applied field of 1 kG, Fig. 5(a) shows no evidence of a first order transition for  $x > 0$ . However,  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$  has evidence for a first order phase transition as found in  $\rho(T)$  at zero field. The inset to Fig. 6(a) shows the electrical resistance along the  $a$  axis in which a jump in the resistance upon warming at  $T_{C2}=21$  K is found. In addition, one of the properties of  $\text{SmCrSb}_3$  below  $T_{C2}$  is that, along the  $b$  axis at 5 K,  $M(H)$  has a large hysteresis loop which is entirely contained within the first quadrant. Fig. 6(a) shows the magnetization along the  $b$  axis for  $\text{Sm}_{1-x}\text{La}_x\text{CrSb}_3$ , which shows that this off-center hysteresis loop persists for  $x = 0.22$ . The rounded nature of this loop is most likely due to misalignment of the sample. Increasing the concentration of  $\text{La}^{3+}$  reduces the width of the hysteresis loop, and moves its position so that it is centered about zero field, as one would expect for a FM phase.

The first order phase transition which is only found in the pure compounds for  $\text{SmCrSb}_3$  is fairly unstable with respect to doping on the rare-earth site. While  $T_{C2}$  is still sharp with up to 45%  $\text{Gd}^{3+}$ , no hysteresis is observed, and the transition is no longer first order. The only alloy in which the transition continues to be first order is for 22%  $\text{La}^{3+}$ , which has hysteresis with zero magnetic field in the electrical resistivity. In addition,  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$  shares the signature for this unique phase with an off-center magnetization vs field loop contained primarily in the first quadrant.

$\text{PrCrSb}_3$  has FM ordering along the  $b$  axis due to the itinerant  $3d$ -electrons at  $T_{C1}=111$  K, followed by FM ordering of the localized  $4f$ -electrons at  $T_{C2}=18$  K along the  $a$  axis. These

magnetic properties are very similar to  $\text{NdCrSb}_3$ , [7–9] which has been investigated with neutron diffraction measurements. [10] Fig. 7(b) shows  $M(T)$  along the  $b$  axis in which a strong increase at  $T_{C1}$  due to the Cr sub-lattice ordering with its easy axis along this direction, followed by a drop below  $T_{C2}$  due to a change in the direction of the easy axis accompanied with the FM ordering of the localized  $\text{Pr}^{3+}$  moments. A dramatic increase in the magnetization along the  $a$  axis at  $T_{C2}$  is due to the new easy axis being oriented in this out-of-plane direction.

Doping with  $\text{La}^{3+}$  removes  $4f$ -moments from the system, and the rare-earth ordering temperature monotonically decreases to zero (Fig. 7(a)). The sharp increase in the magnetization below  $T_{C2}$  also decreases in magnitude. Conversely, doping with  $\text{Gd}^{3+}$  adds  $4f$ -electrons to the system, and  $T_{C2}$  increases to higher temperatures (Fig. 7(b)).  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$  shows the presence of a strong first order phase transition at 16 K, with a hysteresis loop approximately 6 K wide which is similar to the transition found for  $\text{SmCrSb}_3$ . In addition, the unique off-center  $M(H)$  hysteresis loop which is associated with this phase for  $\text{SmCrSb}_3$  also appears for  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$  (Fig. 6(b)). Increasing the concentration of  $\text{Gd}^{3+}$  beyond 21% shows no evidence for a FM ground state.

$\text{PrCrSb}_3$  has two distinct magnetically ordered phases, but neither of these are associated with a first order phase transition. However,  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$  has a first order phase transition at  $T_{C2}=16$  K, which shares several properties with  $\text{SmCrSb}_3$ . Among these are the sharp drop in  $M(T)$  along the  $b$  axis, and hysteresis in temperature as well as in  $M(H)$  in which the hysteresis loop is almost entirely contained within the first quadrant. Finally, the FM ground state is suppressed with  $\text{Gd}^{3+}$  doping, but persists for  $\text{La}^{3+}$  doping. Only in this case, the easy axis at 5 K changes from the  $a$  axis to the  $b$  axis (data not shown).

#### IV. DISCUSSION

The magnetism of rare-earth compounds is often described by the Ruderman-Kittel-Kasuya-Yosida (RKKY) interaction, in which the localized rare-earth moments interact through an indirect exchange via conduction electrons. The transition temperature,  $\theta$ , within the Weiss molecular field theory, is related to the de Gennes factor,  $DG = (g - 1)^2 J(J + 1)$ ,

where  $g$  is the Landé  $g$ -factor, and  $J$  is the total spin angular momentum, by,[15]

$$\theta = 2\pi z A_0^2 / k_B N(E_F) DG \sum_{\mathbf{R}_i \neq \mathbf{R}_j} \phi(2k_F |\mathbf{R}_i - \mathbf{R}_j|), \quad (1)$$

in which there are  $z$  conduction electrons per atomic volume,  $A_0$  is the first order coupling between the localized spins and the conduction electrons,  $N(E_F)$  is the density of states at the Fermi level,  $k_F$  is the Fermi wave number,  $\mathbf{R}_i$  is the lattice site of the  $i^{\text{th}}$  localized spin, and  $\phi(x) = (\sin(x) - x \cos(x))/x^4$  is an oscillatory function due to the wave number dependent susceptibility. Eq. 1 shows that the magnetic ordering temperature should be linearly related to the de Gennes factor, and the type of ordering (FM or AFM) depends upon the oscillatory function,  $\phi(x)$ .

Fig. 8 shows both  $T_{C1}$ , due to the itinerant Cr ions, and  $T_{C2}$ , due to the localized rare-earth ions. This figure combines all of the pure compounds,  $R\text{CrSb}_3$  ( $R = \text{La-Nd, Sm, Gd}$ )[3, 12, 13, 16], as well as the alloys investigated in this study. All of the rare-earth ordering temperatures linearly increase with the de Gennes factor at  $dT_{C2}/d(DG) = 5$  K, while the FM ordering due to the Cr ion decreases with de Gennes factor at a rate of  $dT_{C1}/d(DG) = -2$  K. The two magnetic ordering phases converge to a single value at  $\text{GdCrSb}_3$ . This provides support to the conclusion that the single magnetic ordering temperature in  $\text{GdCrSb}_3$  is a ferrimagnetic phase in which the itinerant  $3d$  sub-lattice and the localized  $4f$  sub-lattices are anti-aligned. The linear relationship between the rare-earth ordering temperature and de Gennes factor indicates that the interactions between the  $4f$  moments are due to the RKKY interaction.

Alloying with  $\text{Gd}^{3+}$  not only varies the de Gennes factor by increasing the number of  $4f$ -electrons, but it also decreases the unit cell volume because  $\text{Gd}^{3+}$  is smaller than  $\text{Sm}^{3+}$ . It is therefore useful to compare these results with that of applying physical pressure to the system.  $\text{SmCrSb}_3$  has a large increase in the electrical resistivity along the  $a$  axis at the first order phase transition.[12] It is therefore ideally suited to investigate the effects of pressure on the rare-earth ordering, and the results are shown in Fig. 9. It is immediately clear that  $\text{Gd}^{3+}$  doping has the opposite effect on  $T_{C2}$  from hydrostatic pressure on  $\text{SmCrSb}_3$ , illustrating that the volume change is not the primary cause for the results presented here.

The moments at 55 kG and 5 K for the alloys are shown in Fig. 10. If the magnetization has saturated, this plot shows the value as a solid marker. The reason for the distinction is that several of the magnetization curves continue to increase at the maximum attainable

field of 55 kG. The very low value of magnetization for  $\text{Pr}_{0.61}\text{Gd}_{0.39}\text{CrSb}_3$  ( $DG = 6.63$ ) along the  $b$  axis, for example, is due to this alloy being in a non-FM ground state (see Fig. 7(b)) in which the magnetization gradually increases with field (Fig. 6(b)). The data generally indicate a minimum in the magnetization near  $DG \simeq 4.5$ , and no axis has saturated by 55 kG for  $DG > 5$ . All of the FM ground states are found for  $DG < 4$ , and ferrimagnetic ground states for  $DG > 4.5$ .

In the paramagnetic phase ( $200 \text{ K} \leq T \leq 350 \text{ K}$ ), the magnetic susceptibility can be fit to a modified Curie-Weiss law,  $\chi = C/(T - \theta) + \chi_0$ , where  $C$  is the Curie constant and is related to the effective moment,  $\theta$  is the paramagnetic Weiss temperature and indicates the strength and sign of the interaction, and  $\chi_0$  is a temperature independent constant. The average value of the paramagnetic Weiss temperature,  $\theta$ , obtained from the three principle axes for each alloy is shown in Fig. 11. It is clear that two distinct regions exist, centered near  $\text{SmCrSb}_3$  ( $DG \sim 4.5$ ). First, for  $DG < 3.5$  the Weiss temperature is positive, indicative of FM interactions, and falls at a slow rate of  $d\theta/d(DG) = -3 \text{ K}$ . This is very close to the decrease found in the Curie temperature,  $dT_{C1}/d(DG) = -2 \text{ K}$ , which shows that the ground state is dominated by the FM Cr sub-lattice. However, for  $DG > 4.5$ , the Weiss temperature linearly decreases at a rate five times faster,  $d\theta/d(DG) = -16 \text{ K}$ , and goes from positive (FM coupling) to negative (AFM coupling) values. In this region, the strength of the rare-earth coupling has a large effect on the ground state of the system, and the anti-alignment of the sub-lattices becomes increasingly dominant.

The shaded region near  $DG \simeq 4$  in Fig. 11 contains the results of four alloys, three of which exhibit a first order phase transition to an AFM phase ( $\text{SmCrSb}_3$ ,  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$ , and  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$ ). While  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  did not have a first order transition ( $DG = 4.1$ ,  $x = 0.25$  fell into this region but the transition at  $T_{C2}$  was not first order), this is likely due to the fact that  $\text{Gd}^{3+}$  has no orbital angular momentum, so that the magnetic anisotropy of this ion is greatly reduced. This shaded region is also the crossover from a FM ground state in which the localized rare-earth moments and the itinerant Cr sub-lattice align with the same easy axis for  $DG < 3.5$ , and a ferrimagnetic ground state in which the rare-earth magnetic sub-lattice and the Cr sub-lattice anti-align with each other. Both Cr and Sb vacancies are known to exist in these family of compounds,[1, 4, 10] which appears to become more important as the rare-earth size decreases.  $\text{YbCrSb}_3$  has also been shown to be the only member in which the rare-earth is divalent[14] due to electron transfer from the

Cr-Sb octahedral to the Yb-Sb checkerboard plane. Similar changes in stoichiometry may transfer electrons from the valence band to bring about the change in coupling between the Cr and rare-earth sub-lattice. Further detailed investigations into the vacancies within the crystal structure will need to be carried out in order to investigate the substoichiometry of the compounds, both in detailed x-ray analysis, and by chemically varying the Cr and Sb content.

## V. CONCLUSIONS

Alloying  $\text{La}^{3+}$  or  $\text{Gd}^{3+}$  into  $R\text{CrSb}_3$  for  $R=(\text{La}, \text{Pr}, \text{Sm}, \text{or Gd})$  has the effect of varying the interactions between the localized  $4f$  moments and the itinerant  $3d$  moments so that the ground state of these materials varies from FM ( $R = \text{La-Nd}$ ) to ferrimagnetic ( $R=\text{Sm}$  and  $\text{Gd}$ ). The linear relationship found between the rare-earth ordering temperature and the de Gennes factor shows that the interaction between the  $4f$  moments is due to the RKKY mechanism. The two magnetic ordering phases converge to a single ordering temperature for  $\text{GdCrSb}_3$ , suggesting that the ordering for this material is ferrimagnetic, in which the Cr sub-lattice anti-aligns with the localized rare-earth moments. The paramagnetic Weiss temperature, on the other hand, has a sharp kink near  $DG \simeq 4.5$  ( $\text{SmCrSb}_3$ ), and the saturation value of the magnetization for the alloys has a minimum near this point. In addition, a first order phase transition is found for  $\text{SmCrSb}_3$  ( $DG= 4.46$ ),  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$  ( $DG = 3.48$ ), and  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$  ( $DG = 3.94$ ), which all lie near this kink. This suggests that the unique first order phase transition found at 30 K for  $\text{SmCrSb}_3$  is due to its location in this phase diagram where the coupling between the  $3d$  and  $4f$  moments change from FM to AFM. As the de Gennes factor continues to increase, AFM coupling between the magnetic sub-lattices becomes dominant and the Weiss temperature goes from an overall positive value to negative. Compounds with  $DG < 3.5$  have an itinerant FM transition due to the Cr sub-lattice. The rare-earth moments also couple with FM interactions, which leads to a FM ground state. Therefore, within this region, the Weiss temperature remains relatively constant and positive. While the Cr sub-lattice continues to undergo FM alignment for all the alloys across the series, the rare-earth moments begin to exhibit AFM coupling with the Cr sub-lattice for  $DG > 4.5$ . This causes the Weiss temperature to drop and eventually turn negative.  $\text{GdCrSb}_3$  is therefore found to have a single ordering temperature in which

the Cr magnetic sub-lattice is anti-aligned with the localized  $\text{Gd}^{3+}$  sub-lattice.

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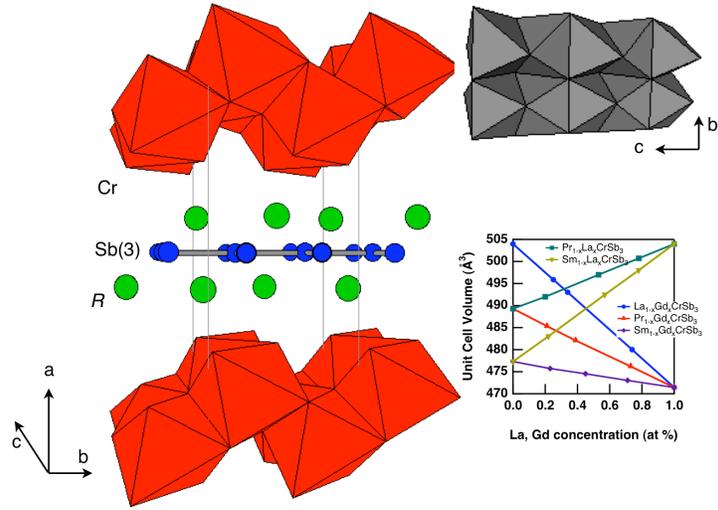


FIG. 1: The crystal structure of  $RCrSb_3$ . The top inset shows the chains of the face-sharing Cr octahedra, while the bottom inset shows the unit cell volume of the alloys in the current investigation.

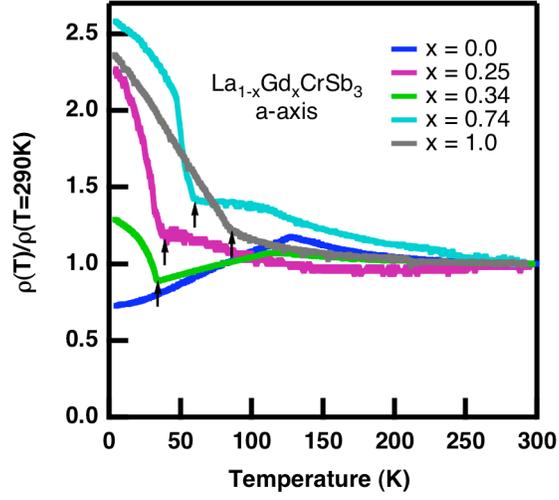


FIG. 2: Normalized resistance vs temperature along the  $a$  axis for  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$ . The arrows point to the rare-earth ordering temperature,  $T_{C2}$ .

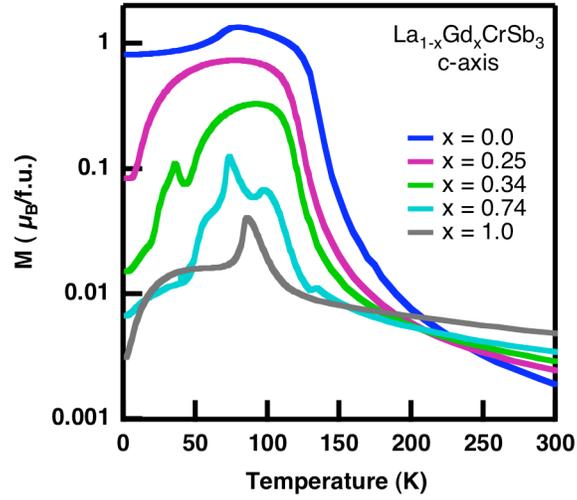


FIG. 3: Magnetization vs temperature along the  $c$  axis for  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  with an applied field of 1 kG. The data is shown on a semi-log plot in order for the magnetization transition to be seen for each alloy.

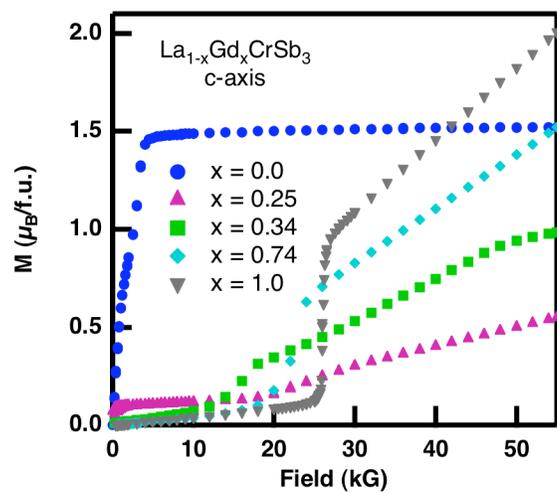


FIG. 4: Magnetization vs field along the  $c$  axis for  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$  at 5 K.

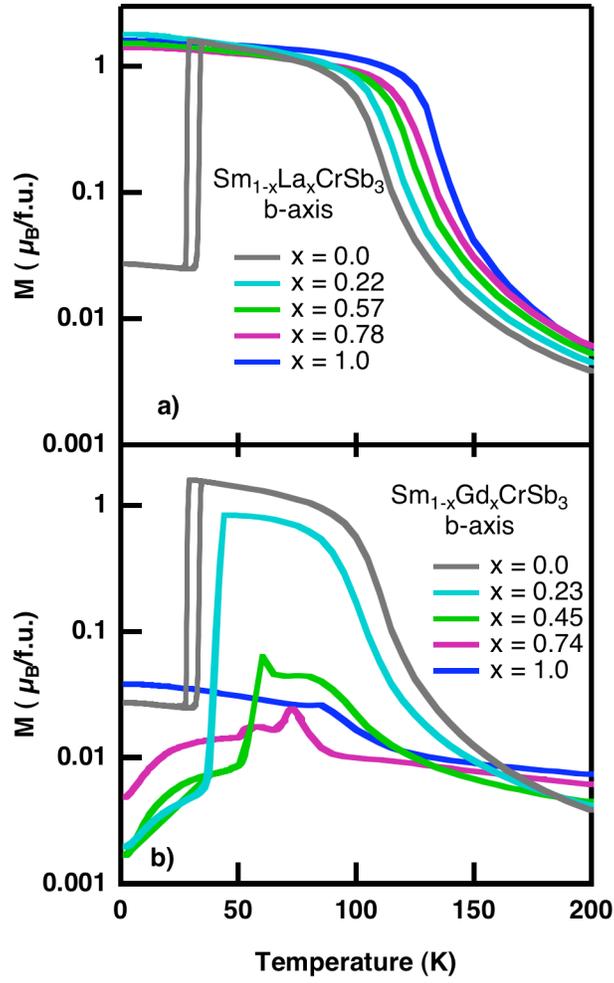


FIG. 5: A semi-log plot of the magnetization vs temperature along the  $b$  axis for a)  $\text{Sm}_{1-x}\text{La}_x\text{CrSb}_3$  and b)  $\text{Sm}_{1-y}\text{Gd}_y\text{CrSb}_3$  with an applied field of 1 kG. The hysteresis due to the first order phase transition is located at  $\text{SmCrSb}_3$  at  $T_{C2}=30$  K.

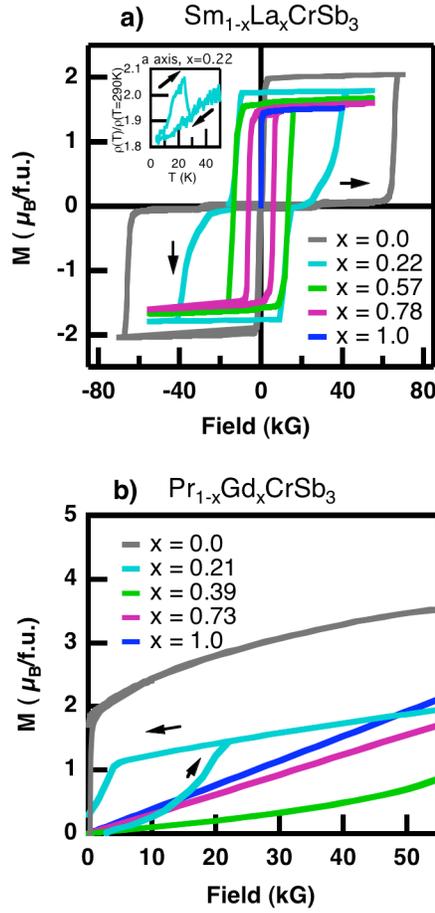


FIG. 6: Magnetization vs field along the  $b$  axis for a)  $\text{Sm}_{1-x}\text{La}_x\text{CrSb}_3$  at 5 K and b)  $\text{Pr}_{1-x}\text{Gd}_x\text{CrSb}_3$ . The inset to a) shows the normalized electrical resistivity along the  $a$  axis for  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$  with the  $H = 0$  hysteresis loop. The off-center hysteresis loop can be seen for  $\text{SmCrSb}_3$ ,  $\text{Sm}_{0.78}\text{La}_{0.22}\text{CrSb}_3$ , and  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$ .

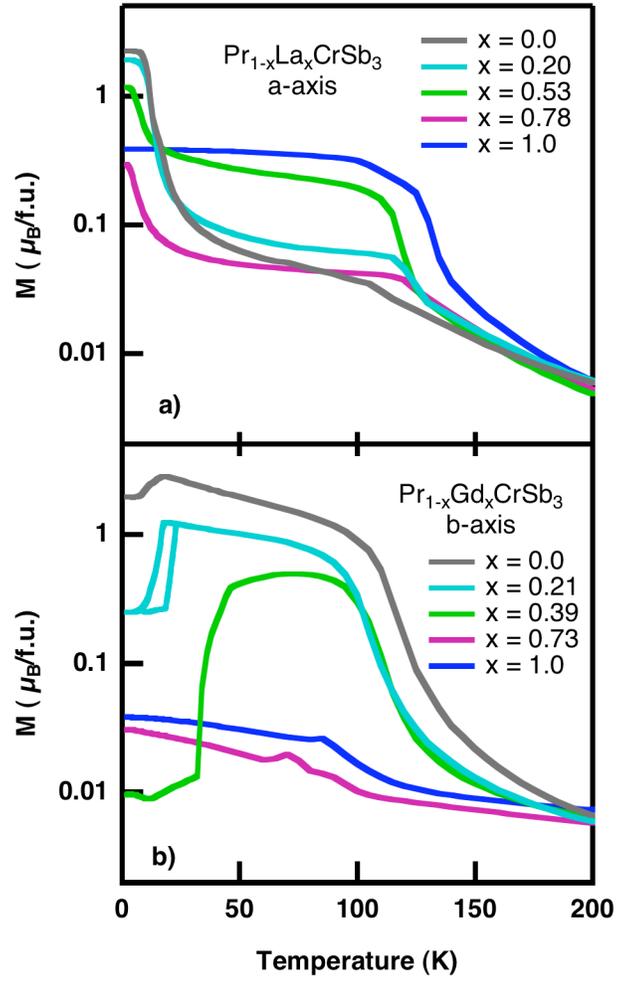


FIG. 7: A semi-log plot of the magnetization vs temperature for a)  $\text{Pr}_{1-x}\text{La}_x\text{CrSb}_3$  along the  $a$  axis, and b)  $\text{Pr}_{1-y}\text{Gd}_y\text{CrSb}_3$  along the  $b$  axis with an applied field of 1 kG. A first order phase transition is found for  $\text{Pr}_{0.79}\text{Gd}_{0.21}\text{CrSb}_3$  at  $T_{C2}=16$  K.

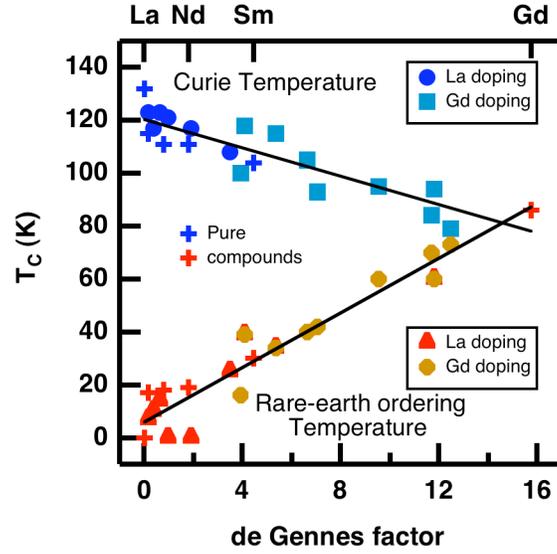


FIG. 8: A plot of the Curie temperature and the rare-earth ordering temperature vs de Gennes factor for the pure compounds,  $RCrSb_3$  ( $R = \text{La-Nd, Sm, Gd}$ ), as well as the La and Gd doped samples investigated here. The top axis shows the location of the rare-earth elements associated with the de Gennes factor, plotted on the bottom axis.

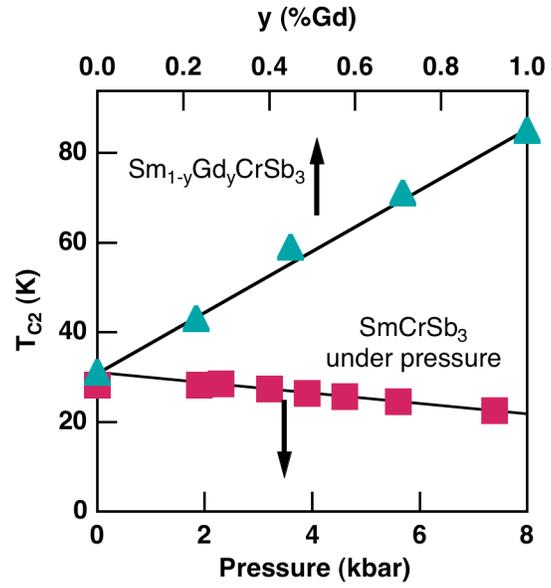


FIG. 9: A comparison between the Néel temperature of  $\text{Sm}_{1-x}\text{Gd}_x\text{CrSb}_3$  (triangles, top axis) and  $\text{SmCrSb}_3$  under pressure (squares, bottom axis).

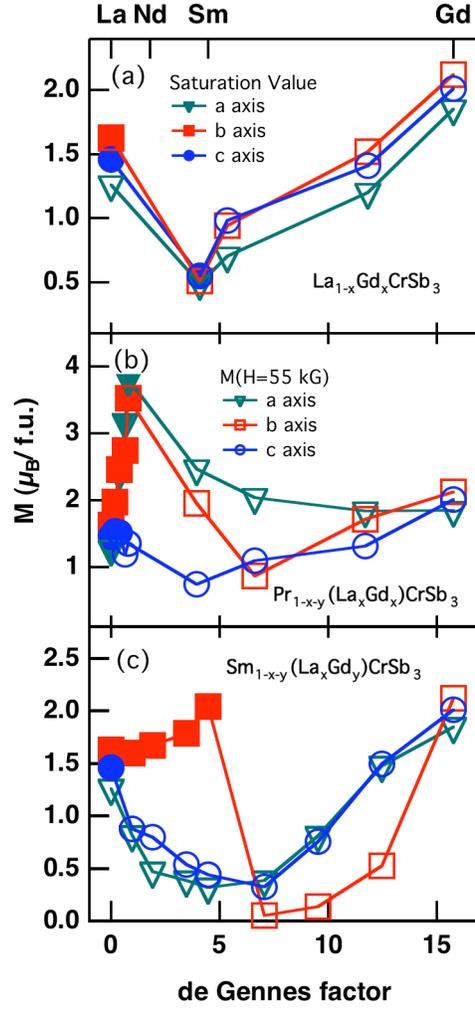


FIG. 10: Magnetization at 5 K (saturated values represented by closed symbols) for the  $a$  axis (triangles),  $b$  axis (squares), and  $c$  axis (circles) for (a)  $\text{La}_{1-x}\text{Gd}_x\text{CrSb}_3$ , (b)  $\text{Pr}_{1-x-y}(\text{La}_x\text{Gd}_y)\text{CrSb}_3$ , and (c)  $\text{Sm}_{1-x-y}(\text{La}_x\text{Gd}_y)\text{CrSb}_3$ . Solid symbols represent the saturation value, and hollow symbols represent the value reached at 55 kG. The top axis shows the location of the rare-earth elements associated with the de Gennes factor, plotted on the bottom axis.

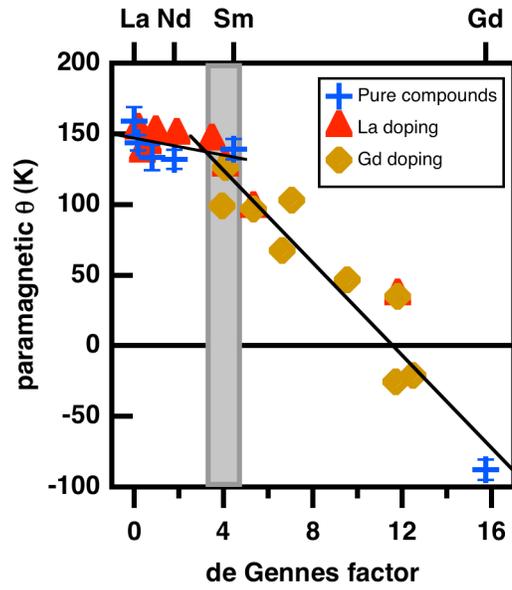


FIG. 11: Value of  $\theta$  obtained from the average of the three principle axis in the paramagnetic phase,  $T > 200$  K for the pure compounds (crosses), the La doped compounds (triangles), and Gd doped samples (diamonds). The shaded regions shows the region of the 1st order AFM transition. The top axis shows the location of the rare-earth elements associated with the de Gennes factor, plotted on the bottom axis.